

# Li<sup>+</sup> Ion Diffusion in Solid State Electrolyte Li<sub>3</sub>InCl<sub>6</sub> measured by <sup>7</sup>Li Liquid State NMR

**Sarah E. Mailhiot<sup>1\*</sup>, Palanivel Molaiyan<sup>1</sup>, Anu M. Kantola<sup>1</sup>, Tao Hu<sup>1</sup>, Ulla Lassi<sup>1</sup>, Ville-Veikko Telkki<sup>1</sup>,**

<sup>1</sup>University of Oulu, Oulu, Finland

\*sarah.mailhiot@oulu.fi

Conventional lithium-ion batteries have changed modern life. However, concerns remain about their energy density limit and safety risks introduced by flammable liquid electrolytes. One alternative is all-solid-state Li batteries (ASSBs) which are not only safer but also can achieve higher energy densities. One of the most vital components of ASSBs is the solid-state electrolyte (SSEs). The optimization and design of SSEs is still underway, of which ion conductivity, Li<sup>+</sup> diffusion mechanisms, and their stability are key. For the study of ion conduction mechanisms and Li<sup>+</sup> diffusion, pulsed gradient spin echo (PGSE) NMR methods are ideal. The PGSE method provides diffusion time dependent <sup>7</sup>Li apparent diffusion coefficients ( $D$ ) and distances on the scale of 0.2-10  $\mu\text{m}$ . In the short diffusion time limit, the electrical conductivity is directly proportional to  $D$  via the Nernst-Einstein equation,

$$\sigma = \frac{DNe^2}{kT}, \quad (1)$$

in which  $\sigma$  is the electrical conductivity,  $N$  is the number of carrier ions,  $e$  is the elementary charge,  $k$  is the Boltzmann constant, and  $T$  is the temperature. In the long diffusion time limit,  $D$  scales with tortuosity and restricted motion, such as the effect of grain boundaries, and the mean squared displacement can be calculated as

$$\langle z^2 \rangle = 6D\Delta \quad (2)$$

in which  $\Delta$  is the diffusion time. In this study, the Li<sup>+</sup> diffusion mechanisms in Li<sub>3</sub>InCl<sub>6</sub> are studied as a function of treatment condition. Li<sub>3</sub>InCl<sub>6</sub> is a halide-based SSEs with a stable monoclinic crystal structure and good ionic conductivity. It is known that the ionic conductivity increases after annealing; however, the mechanism by which conductivity increases is unknown. The results indicate that the SSEs is stable when exposed to air and that the Li<sup>+</sup>  $D$  is enhanced by annealing while the presence of intraparticle restrictions increase. The Li<sub>3</sub>InCl<sub>6</sub> samples are also characterized by  $D$ - $T_2$ ,  $T_2$ ,  $T_2$ - $T_2$ , and  $T_1$  measurements as well as FESEM, TEM/STEM and XRD.

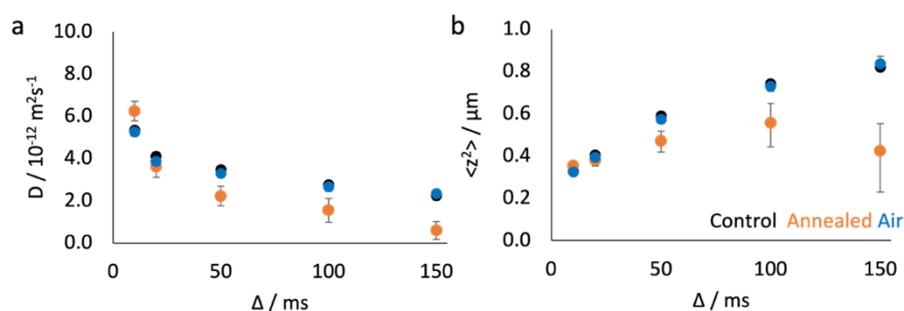


Figure 1: a)  $D$  is plotted as a function of diffusion time ( $\Delta$ ). b) The mean squared displacement,  $\langle z^2 \rangle$  is plotted as a function of  $\Delta$ . When  $\Delta$  is short ( $\Delta = 10$  ms),  $D$  reflects conductivity. In comparison to the control sample, <sup>7</sup>Li  $D(\Delta = 10$  ms) increases after annealing and shows no change when the sample is exposed to air. When  $\Delta$  is long, ( $\Delta = 150$  ms),  $D$  reflects long range mobility and tortuosity. In comparison to the control sample, <sup>7</sup>Li  $D(\Delta = 150$  ms) decreases after annealing and shows no change when the sample is exposed to air.